

## Electrochemical Reactivity of $\text{LiFePO}_4$ Prepared by Hydrothermal Method

Kaoru Dokko,<sup>1,2</sup> Shohei Koizumi,<sup>1</sup> and Kiyoshi Kanamura<sup>\*1,2</sup>

<sup>1</sup>Department of Applied Chemistry, Graduate School of Engineering, Tokyo Metropolitan University,  
1-1 Minami-ohsawa, Hachioji, Tokyo 192-0397

<sup>2</sup>CREST, Japan Science and Technology Agency, 4-1-8 Honcho, Kawaguchi 332-0012

(Received December 26, 2005; CL-051573; E-mail: kanamura-kiyoshi@c.metro-u.ac.jp)

$\text{LiFePO}_4$  was prepared by hydrothermal method starting from  $\text{CH}_3\text{COOLi}$ ,  $\text{FeSO}_4$ , and  $(\text{NH}_4)_2\text{HPO}_4$  at  $170^\circ\text{C}$ . The particle size of the obtained  $\text{LiFePO}_4$  was about  $3\ \mu\text{m}$ . The performance of the  $\text{LiFePO}_4$  as a positive electrode material for rechargeable lithium battery was evaluated in an organic electrolyte. The  $\text{LiFePO}_4$  exhibited a discharge capacity of  $145\ \text{mAh g}^{-1}$  at the electrode potential of  $3.5\ \text{V}$  vs  $\text{Li/Li}^+$ .

Phosphates  $\text{LiMPO}_4$  ( $M = \text{Mn, Fe, Co, or Ni}$ ) have been investigated intensively as promising positive electrode materials for lithium batteries.<sup>1–11</sup> Among these materials,  $\text{LiFePO}_4$  is of the lowest cost and highly compatible to the environment.  $\text{LiFePO}_4$  has a highly stable three-dimensional framework due to strong P–O covalent bonds in  $(\text{PO}_4)^{3-}$  polyanion, which prohibits the liberation of oxygen. These characteristics provide an excellent safety under abuse conditions of the batteries.<sup>1–4</sup>  $\text{LiFePO}_4$  can be synthesized using conventional solid-state reactions.<sup>1</sup> However, it is difficult to attain its full capacity because of its low electronic conductivity and slow diffusion of  $\text{Li}^+$  ion in the olivine structure.<sup>1</sup> In order to overcome these weak points, several alternative synthetic methods have been applied to the preparation of  $\text{LiFePO}_4$ . So far, it has been reported that the control of particle size is useful in increasing the charge and discharge capacities of  $\text{LiFePO}_4$ .<sup>4,5</sup> In addition, highly dispersed conductive substances such as carbonaceous materials and some metals in the positive electrode have been reported as being efficient additives.<sup>5–9</sup>

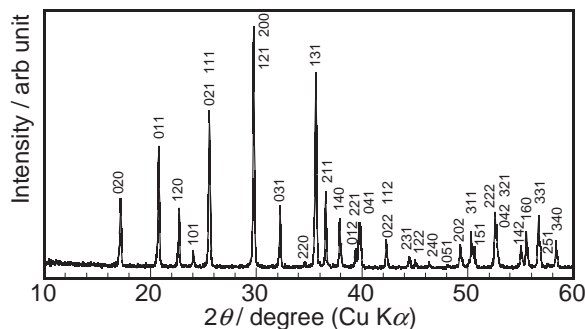
The hydrothermal synthesis is a useful method for preparing fine particles. It also has other advantages such as simple synthesis process and energy efficient.  $\text{LiFePO}_4$  can be prepared by hydrothermal reaction.<sup>12–17</sup> We have prepared  $\text{LiFePO}_4$  from weak basic precursor solutions consisting of  $\text{LiOH}$ ,  $\text{FeSO}_4$ , and  $(\text{NH}_4)_3\text{PO}_4$  by hydrothermal method and reported that an amorphous compound existing on the  $\text{LiFePO}_4$  particle gives a significant influence on its electrochemical reactivity.<sup>14–16</sup> The post-annealing at high temperatures ( $\approx 400^\circ\text{C}$ ) was effective in improving the crystallinity and enhancing the electrochemical reactivity of the hydrothermally synthesized  $\text{LiFePO}_4$ .<sup>14–16</sup> In the present study,  $\text{LiFePO}_4$  was prepared by hydrothermal method using weak acidic precursor solutions at  $170^\circ\text{C}$ , and the electrochemical reactivity of thus prepared  $\text{LiFePO}_4$  without post-annealing was evaluated.

The starting materials were  $\text{CH}_3\text{COOLi}$ ,  $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$ , and  $(\text{NH}_4)_2\text{HPO}_4$ , which were purchased from Wako Pure Chemical Industries.  $\text{CH}_3\text{COOLi}$  and  $(\text{NH}_4)_2\text{HPO}_4$  were dissolved into deionized water, and then aqueous  $\text{FeSO}_4$  solution was added subsequently. In order to prohibit the conversion of  $\text{Fe}^{2+}$  to  $\text{Fe}^{3+}$ , the water was degassed by  $\text{N}_2$  gas bubbling for 30 min prior to preparing the solution, and the mixing process was

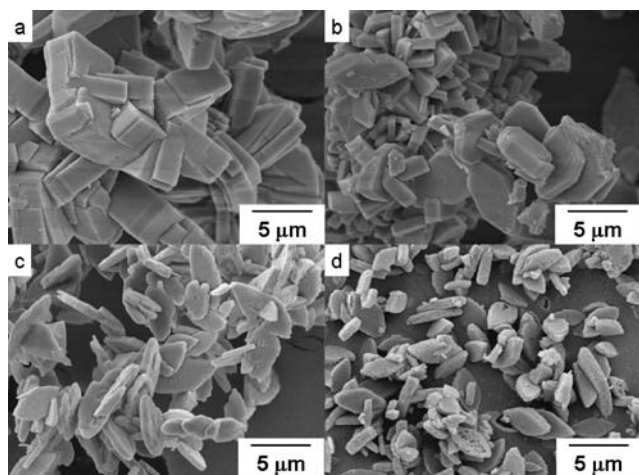
carried out under nitrogen atmosphere. The molar ratio of  $\text{Li}:\text{Fe}:\text{P}$  in the precursor solution was 3:1:1. The concentration of  $\text{Fe}^{2+}$  ( $C_{\text{Fe}}$ ) in the solution was in the range of  $0.2$ – $1.0\ \text{mol dm}^{-3}$ . Forty milliliters of the precursor solution was put into a Teflon vessel (inner volume:  $120\ \text{mL}$ ) sealed in a stainless steel autoclave, and the reactor was heated at  $170^\circ\text{C}$  in an oven for 12 h. During this hydrothermal process, precipitation took place inside the reactor. After the hydrothermal reaction, the reactor was cooled down to room temperature. The precipitated powder was washed with deionized water and acetone. Then, the obtained powder was dried at  $110^\circ\text{C}$  for 1 h under vacuum. The crystalline phases were identified with X-ray diffraction (XRD, RINT-2000, Rigaku) with  $\text{Cu K}\alpha$  radiation. The morphology of the particle was observed by a scanning electron microscope (SEM, JSM-5310, JEOL).

The performance of the  $\text{LiFePO}_4$  as positive electrode was evaluated using a coin-type cell (size 2016) with a lithium metal negative electrode. A composite electrode was prepared by mixing the  $\text{LiFePO}_4$ , Ketjen black, and polytetrafluoroethylene in the weight ratio of 80:15:5. This composite was rolled into a thin sheet with uniform thickness (ca.  $150\ \mu\text{m}$ ), and it was cut into a circle shape ( $13\text{-mm}$  diameter). The mass of  $\text{LiFePO}_4$  contained in the composite positive electrode was  $24.6\ \text{mg}$ . The electrolyte was a mixed solvent of ethylene carbonate and diethyl carbonate (1:1 by volume) containing  $1.0\ \text{mol dm}^{-3}$   $\text{LiClO}_4$ . Galvanostatic charge and discharge tests were performed using an automatic charge–discharge equipment (HJR-110mSM6, Hokuto Denko) in the voltage range of  $2.0$ – $4.2\ \text{V}$  at  $17\ \text{mA g}^{-1}$  ( $0.3\ \text{mA cm}^{-2}$ ). All electrochemical measurements were carried out at  $25^\circ\text{C}$ .

XRD pattern of  $\text{LiFePO}_4$  prepared from the solution of  $C_{\text{Fe}} = 1\ \text{mol dm}^{-3}$  is shown in Figure 1. All the XRD patterns of the prepared materials agreed with that of phospho-olivine  $\text{LiFePO}_4$ ,<sup>18</sup> and no impurity phase was detected. Although all solid products could not be collected, the yields were about 95% for each case. This suggested that  $\text{Fe}^{2+}$  species in the pre-



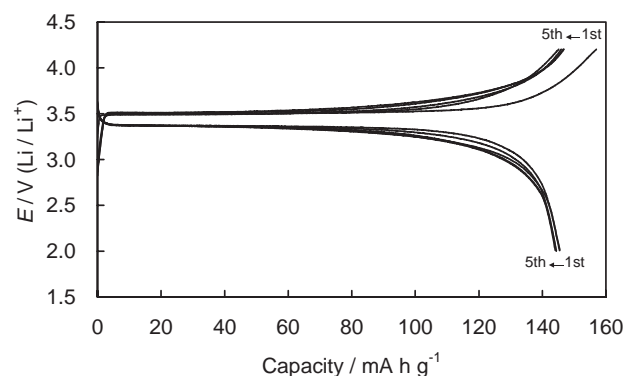
**Figure 1.** XRD pattern of  $\text{LiFePO}_4$  prepared by hydrothermal method from the solution of  $C_{\text{Fe}} = 1.0\ \text{mol dm}^{-3}$ .



**Figure 2.** SEM images of LiFePO<sub>4</sub> prepared by hydrothermal method from the solutions of  $C_{\text{Fe}} = 0.2$  (a), 0.5 (b), 0.8 (c), and 1.0 mol dm<sup>-3</sup> (d).

cursor solutions were mostly converted to LiFePO<sub>4</sub>. The pH value of the precursor solution was ranging from 5.5 to 5.1, and it decreased as increasing the concentration. According to the report by Lee et al., strong basic condition should be avoided to prepare LiFePO<sub>4</sub> without impurity phase.<sup>17</sup> Actually, in our previous work, Fe<sub>2</sub>(PO<sub>4</sub>)OH and Li<sub>3</sub>PO<sub>4</sub> were obtained as impurities in the basic condition (pH  $\approx$  9.5).<sup>15</sup> It is suggested that weak acidic or neutral or weak basic precursor solutions are preferable to obtain single phase of LiFePO<sub>4</sub> by hydrothermal reaction. Figure 2 displays SEM images of the prepared LiFePO<sub>4</sub> powders. The powders were composed of beautiful tabular particles. In fact, the peak intensity ratio in the XRD pattern (Figure 1) is different from that in the literature<sup>11</sup> probably owing to the asymmetric particle texture. The sharp XRD peaks and the beautiful particle shape indicated the high crystallinity of the prepared LiFePO<sub>4</sub>. The particle size of the LiFePO<sub>4</sub> decreased as increasing the concentration of precursor. The particle size of the LiFePO<sub>4</sub> prepared from the solution of  $C_{\text{Fe}} = 1.0$  mol dm<sup>-3</sup> was about 3  $\mu$ m, and the thickness of the particle was ca. 1  $\mu$ m. The powders prepared from the solutions of  $C_{\text{Fe}} = 0.2$  and 0.5 mol dm<sup>-3</sup> were composed of agglomerated particles. On the other hand, highly dispersed particles were obtained from the highest concentrated solution. However, the nucleation and agglomeration mechanisms are not clear at present.

The electronic conductivity of LiFePO<sub>4</sub> is very low, and diffusion of Li<sup>+</sup> ion in the olivine structure is slow.<sup>1</sup> The small particle size, which is helpful for accessibility of the redox centers, is preferable to achieve large discharge capacity.<sup>4,5</sup> Therefore, galvanostatic charge/discharge test was carried out for the smallest LiFePO<sub>4</sub> particles prepared from the solution of  $C_{\text{Fe}} = 1.0$  mol dm<sup>-3</sup>. The charge and discharge curves of hydrothermally synthesized LiFePO<sub>4</sub> are shown in Figure 3. The plateaus were observed in both charge and discharge curves at the potential of 3.5 V vs Li/Li<sup>+</sup>. This electrochemical behavior corresponded to the solid-state redox of Fe<sup>2+/3+</sup> in the LiFePO<sub>4</sub> accompanying with Li<sup>+</sup> ion extraction and insertion.<sup>1</sup> The discharge capacity was 145 mA h g<sup>-1</sup>, which was as high as 85%



**Figure 3.** Charge and discharge curves of LiFePO<sub>4</sub> measured at a constant current of 17 mA g<sup>-1</sup>. LiFePO<sub>4</sub> was prepared by hydrothermal method from the solution of  $C_{\text{Fe}} = 1.0$  mol dm<sup>-3</sup>.

of the theoretical capacity (170 mA h g<sup>-1</sup>).

In conclusion, LiFePO<sub>4</sub> with high electrochemical reactivity was prepared by simple hydrothermal method at low temperature as 170 °C. It was considered that the high electrochemical reactivity was due to the small particle size and high crystallinity.

## References

- 1 A. K. Padhi, K. S. Nanjundaswamy, J. B. Goodenough, *J. Electrochem. Soc.* **1997**, *144*, 1188.
- 2 J. M. Tarascon, M. Armand, *Nature* **2001**, *414*, 359.
- 3 S. Okada, S. Sawa, M. Egashira, J. Yamaki, M. Tabuchi, H. Kageyama, T. Konishi, A. Yoshino, *J. Power Sources* **2001**, *97–98*, 430.
- 4 A. Yamada, S. C. Chung, K. Hinokuma, *J. Electrochem. Soc.* **2001**, *148*, A224.
- 5 H. Huang, S. C. Yin, L. F. Nazar, *Electrochem. Solid-State Lett.* **2001**, *4*, A170.
- 6 N. Ravet, Y. Chouinard, J. F. Magnan, S. Besner, M. Gauthier, M. Armand, *J. Power Sources* **2001**, *97–98*, 503.
- 7 Z. Chen, J. R. Dahn, *J. Electrochem. Soc.* **2002**, *149*, A1184.
- 8 S. T. Myung, S. Komaba, N. Hirosaki, H. Yashiro, N. Kumagai, *Electrochim. Acta* **2004**, *49*, 4213.
- 9 S. Y. Chung, J. T. Bloking, Y. M. Chiang, *Nat. Mater.* **2002**, *1*, 123.
- 10 A. Yamada, M. Yonemura, Y. Takei, N. Sonoyama, R. Kanno, *Electrochem. Solid-State Lett.* **2005**, *8*, A55.
- 11 A. Yamada, H. Koizumi, N. Sonoyama, R. Kanno, *Electrochem. Solid-State Lett.* **2005**, *8*, A409.
- 12 S. Yang, P. Y. Zavalij, M. S. Whittingham, *Electrochem. Commun.* **2001**, *3*, 505.
- 13 S. Yang, Y. Song, P. Y. Zavalij, M. S. Whittingham, *Electrochem. Commun.* **2002**, *4*, 239.
- 14 K. Shiraishi, Y. H. Rho, K. Kanamura, *J. Ceram. Soc. Jpn.* **2004**, *112*, Suppl. 112-1, S58.
- 15 K. Shiraishi, K. Dokko, K. Kanamura, *J. Power Sources* **2005**, *146*, 555.
- 16 K. Dokko, K. Shiraishi, K. Kanamura, *J. Electrochem. Soc.* **2005**, *152*, A2199.
- 17 J. Lee, A. S. Teja, *J. Supercrit. Fluids* **2005**, *35*, 83.
- 18 JCPDS card No. 40-1499.